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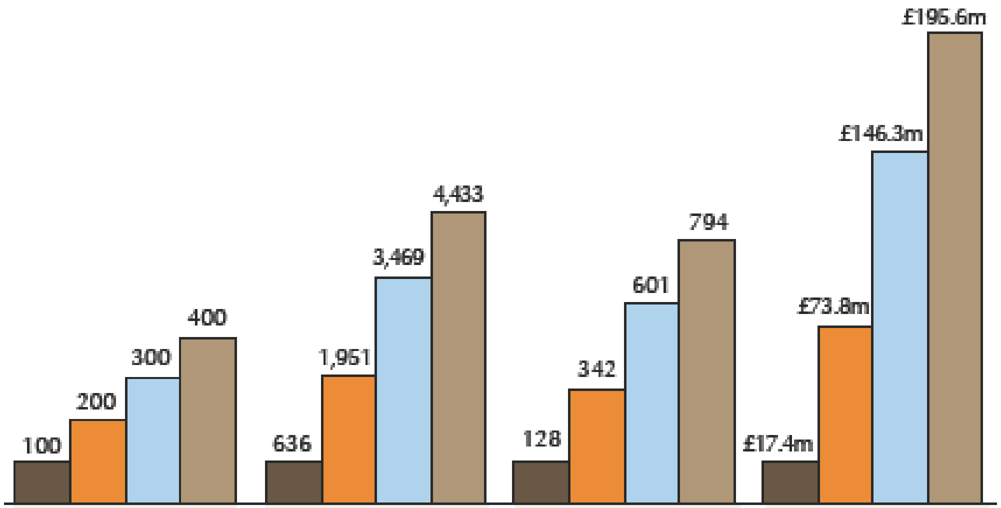

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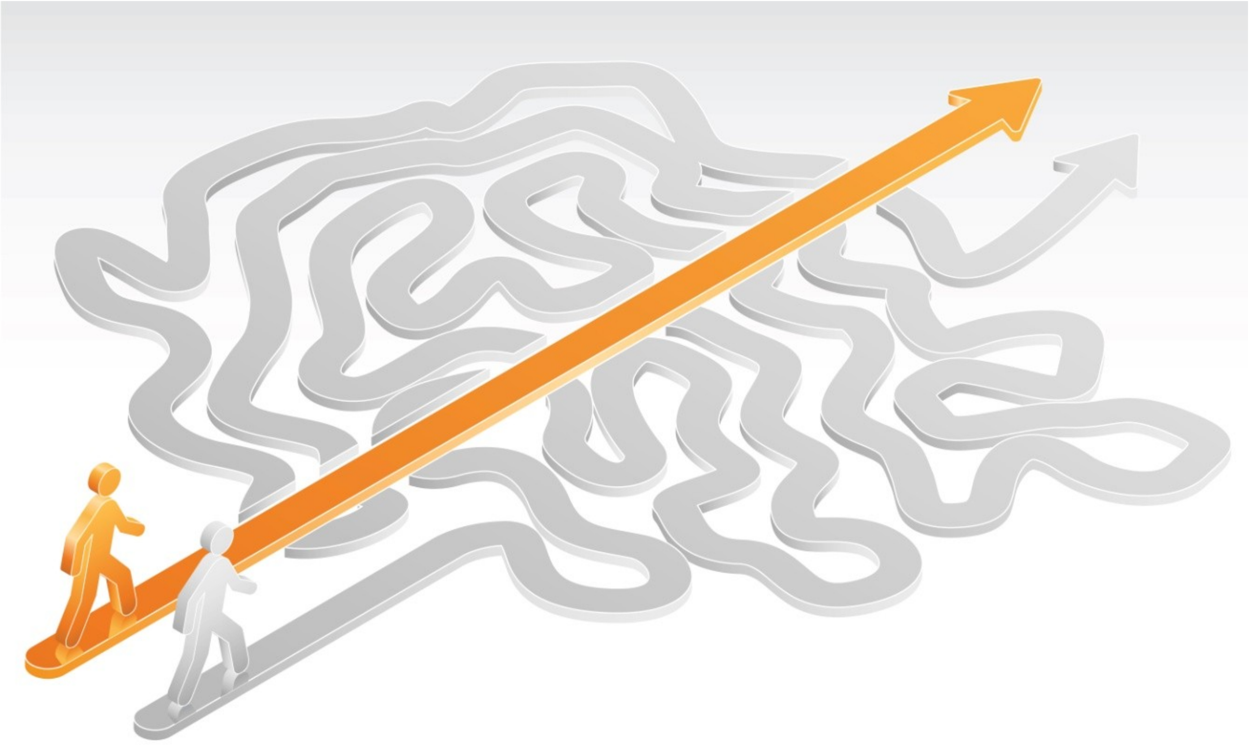

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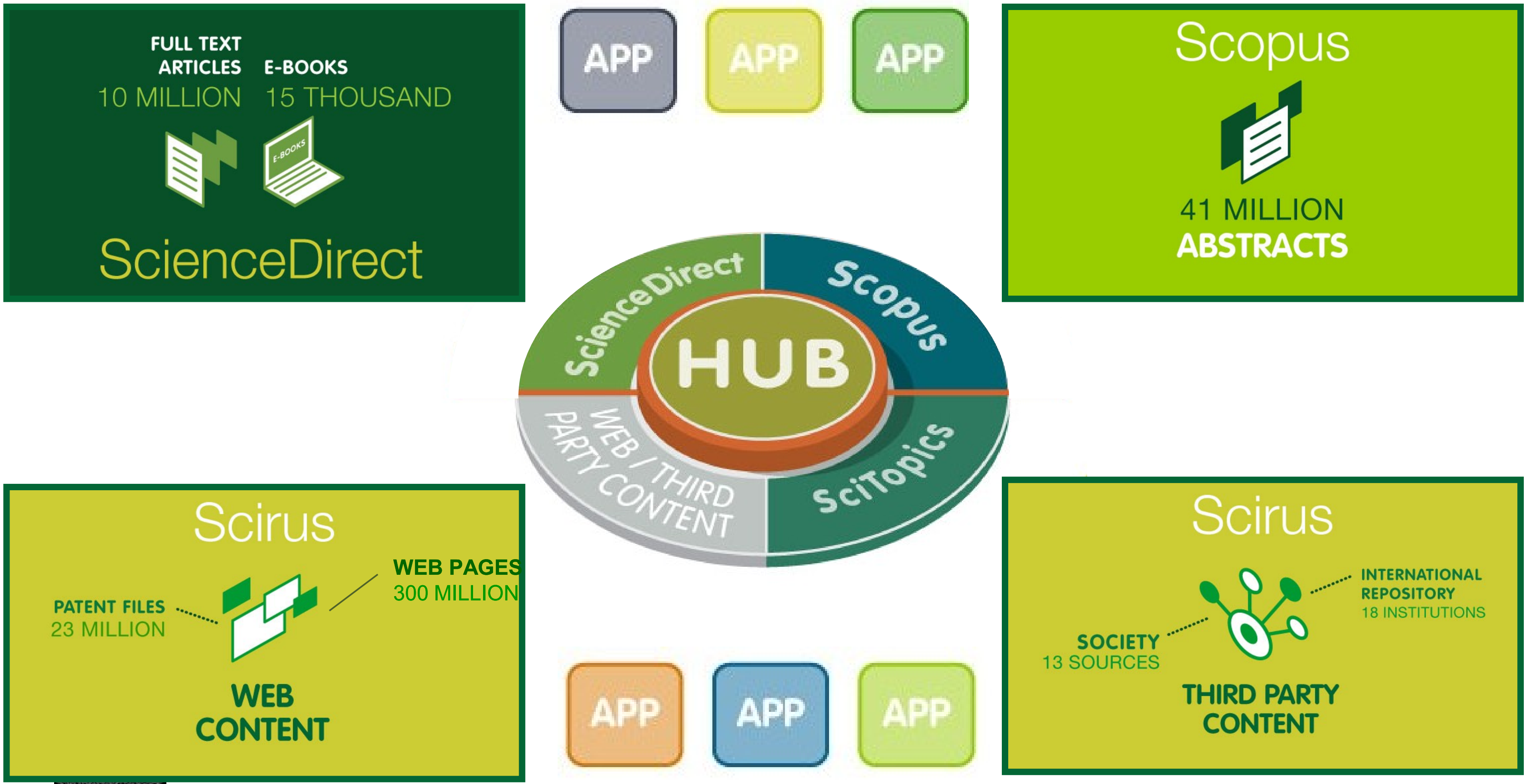

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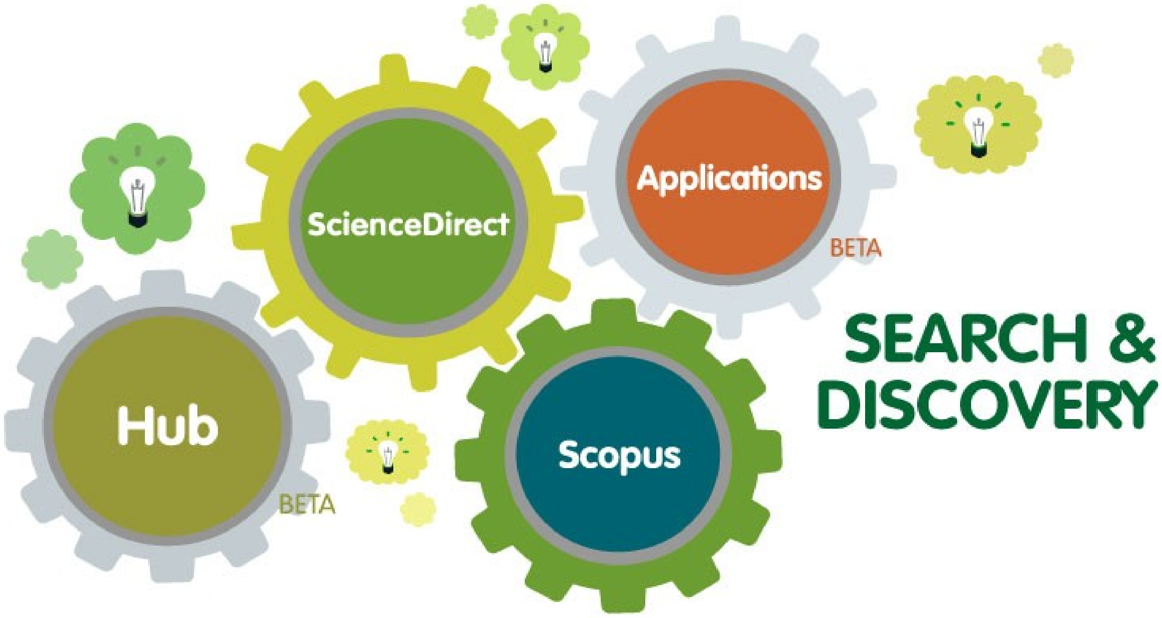

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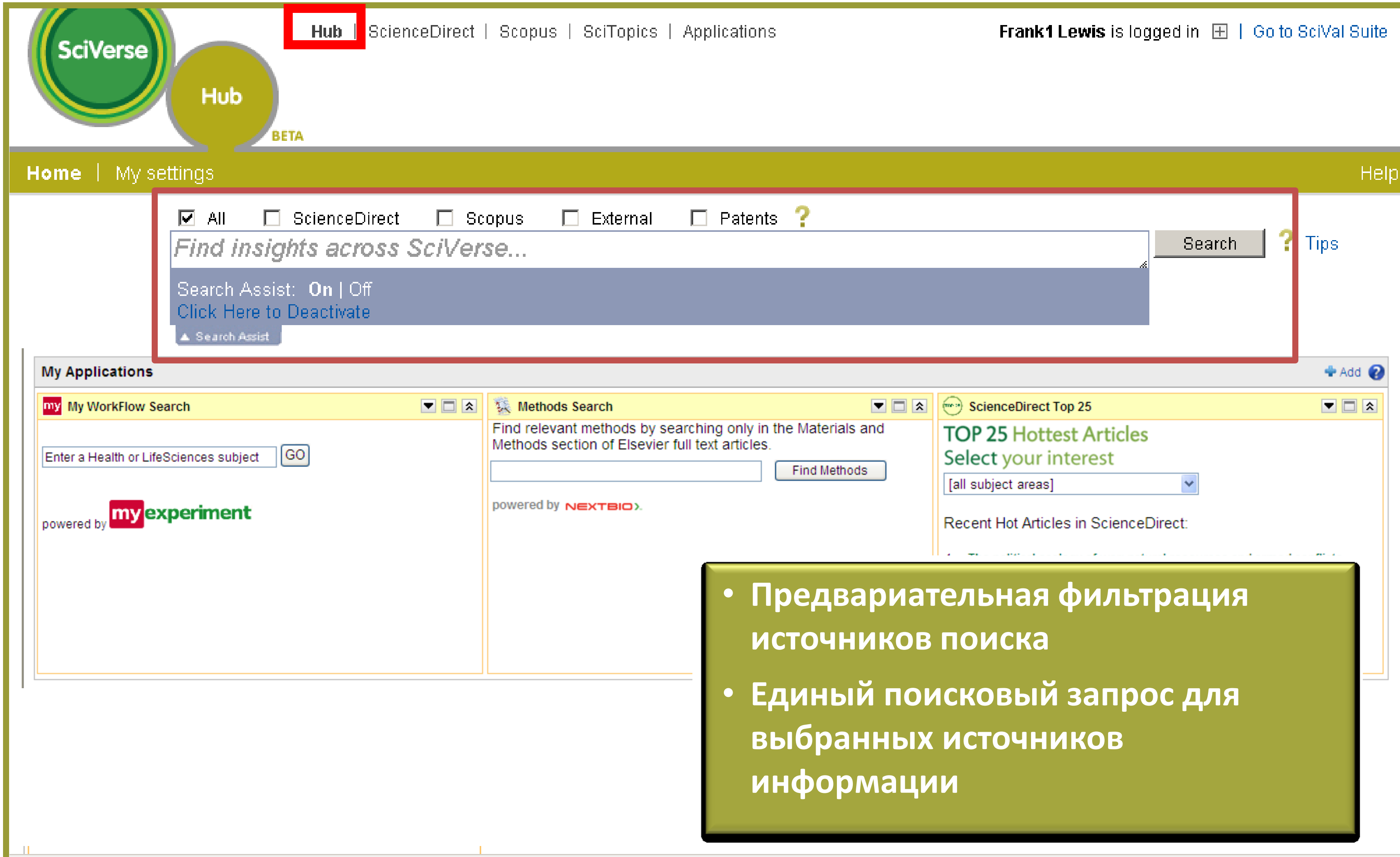

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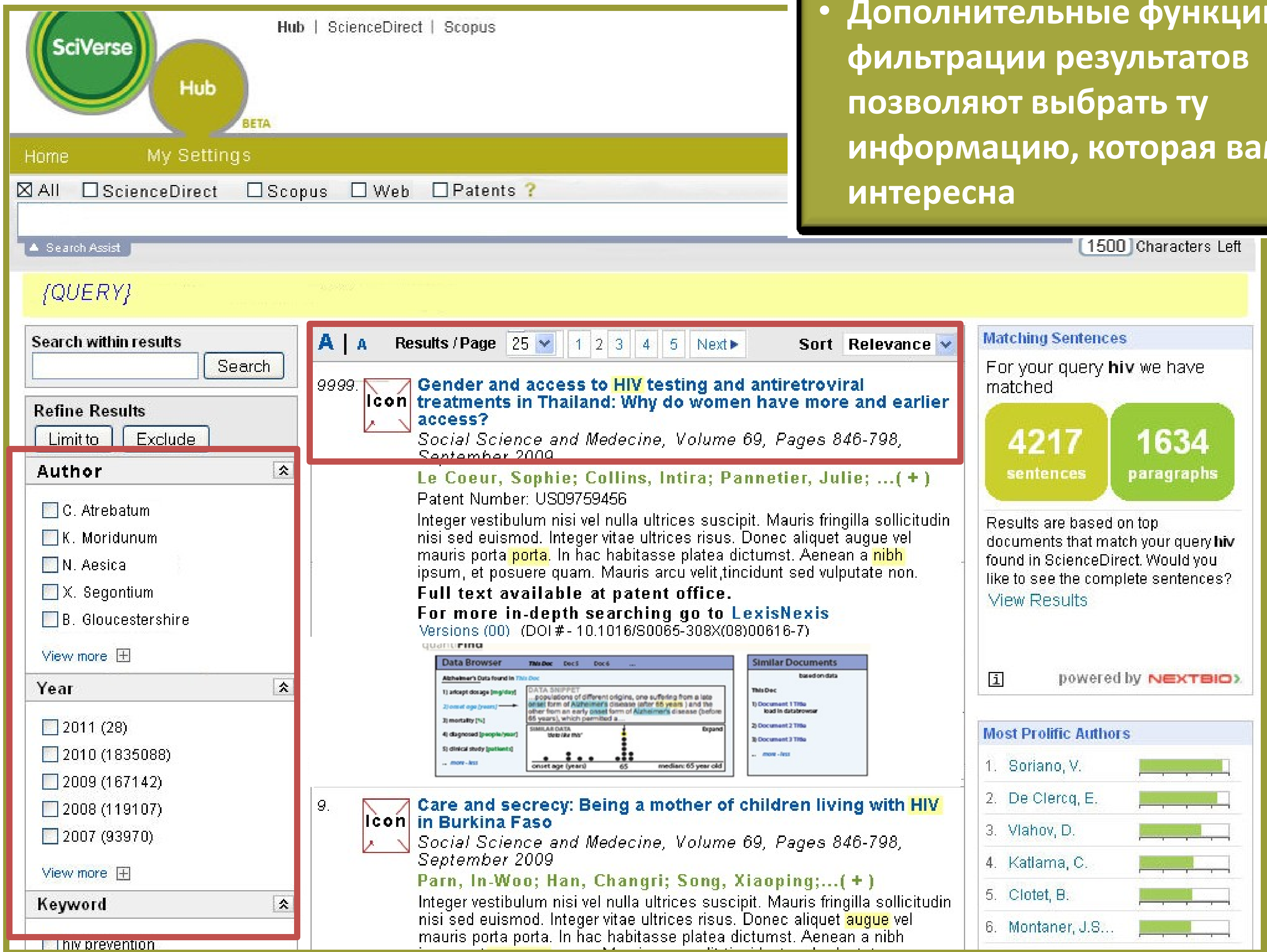

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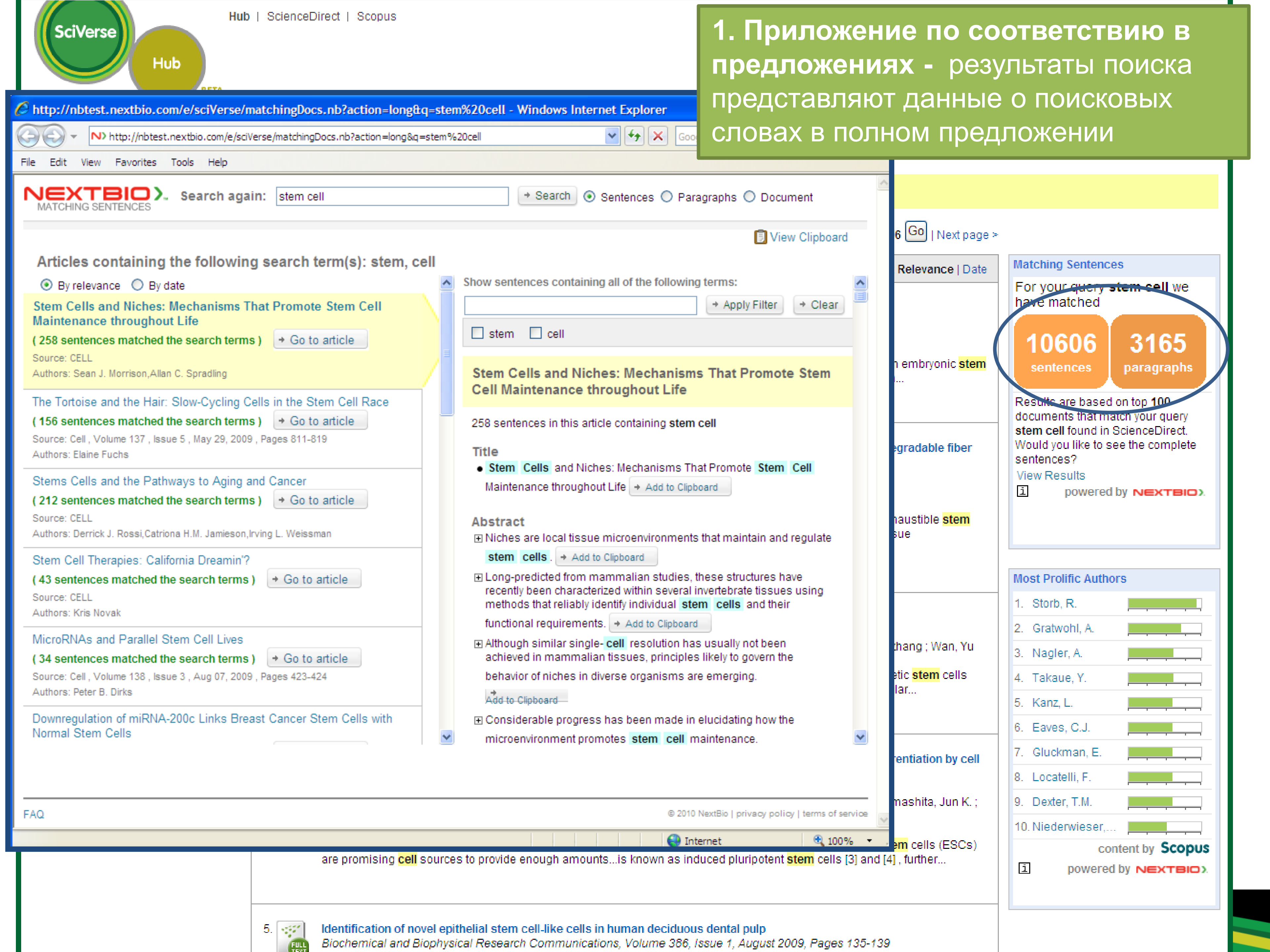

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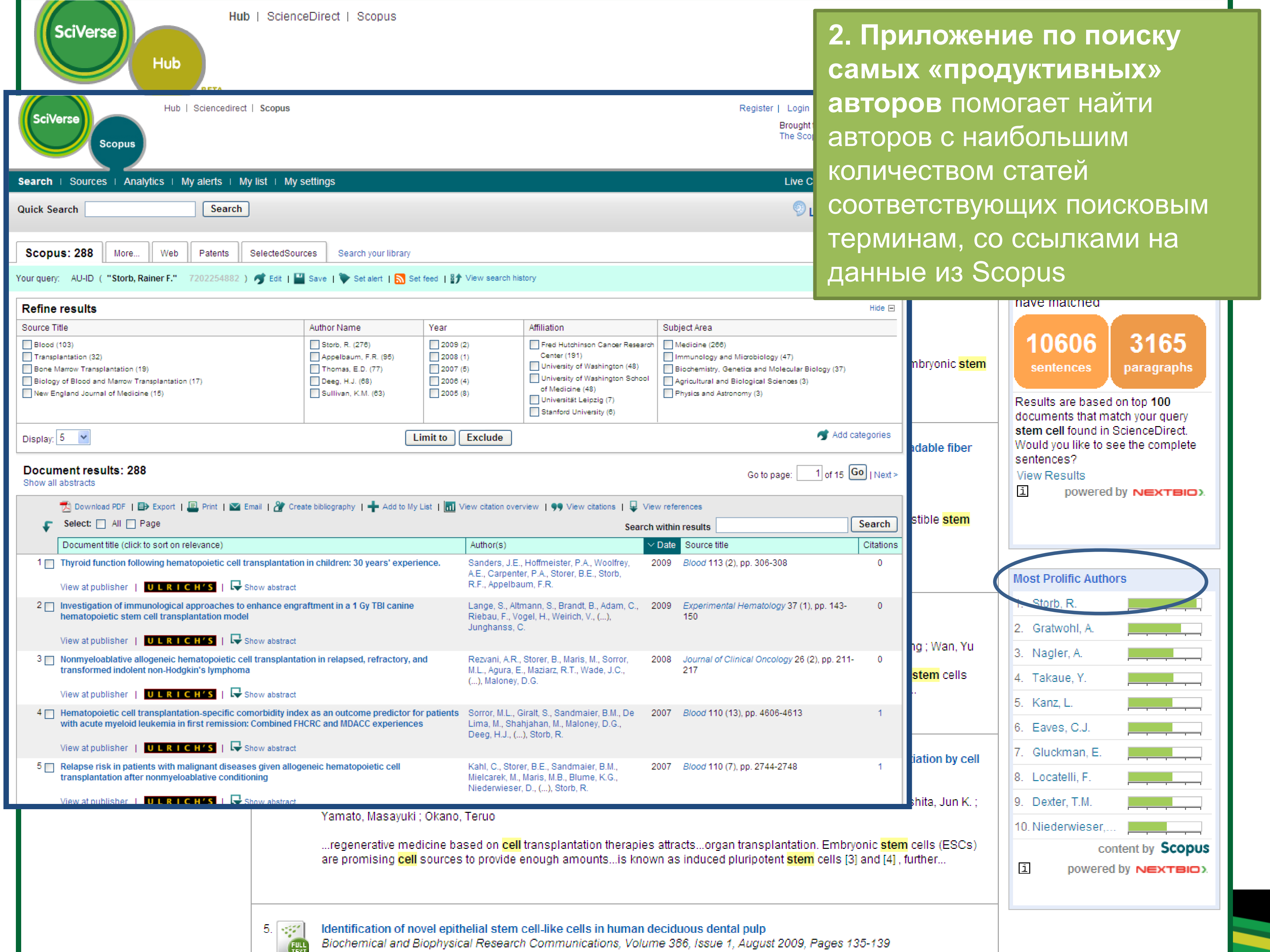

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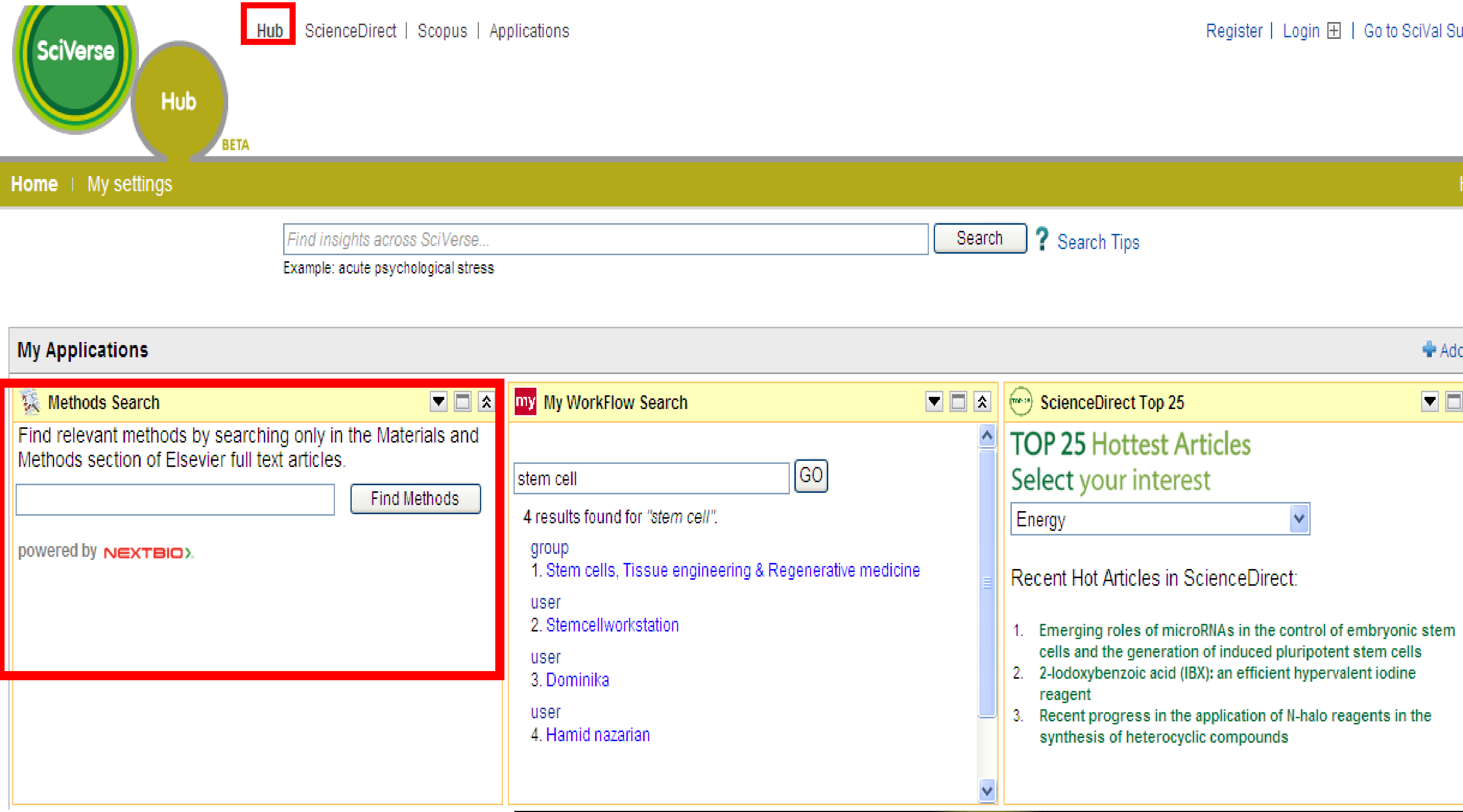

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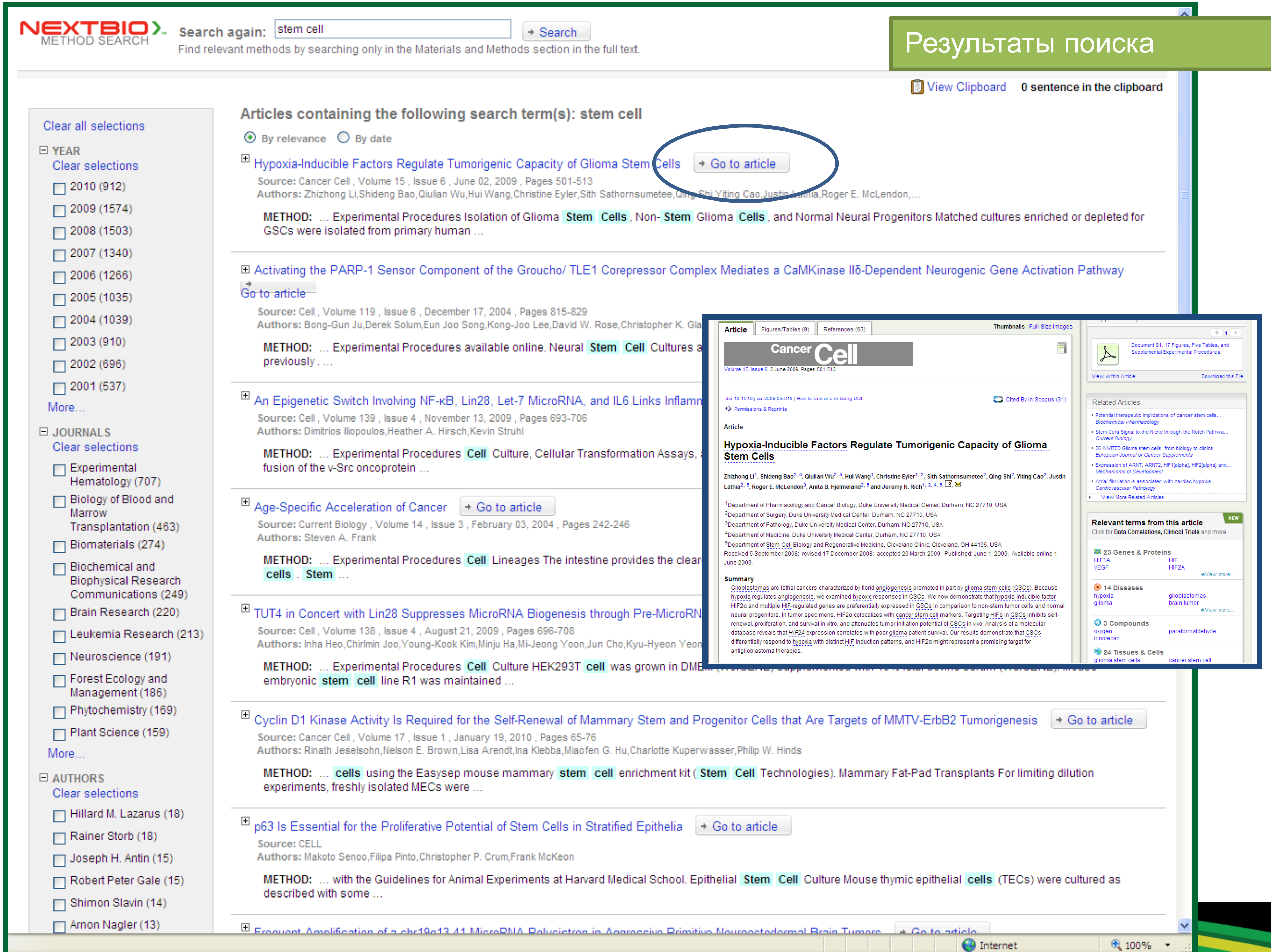

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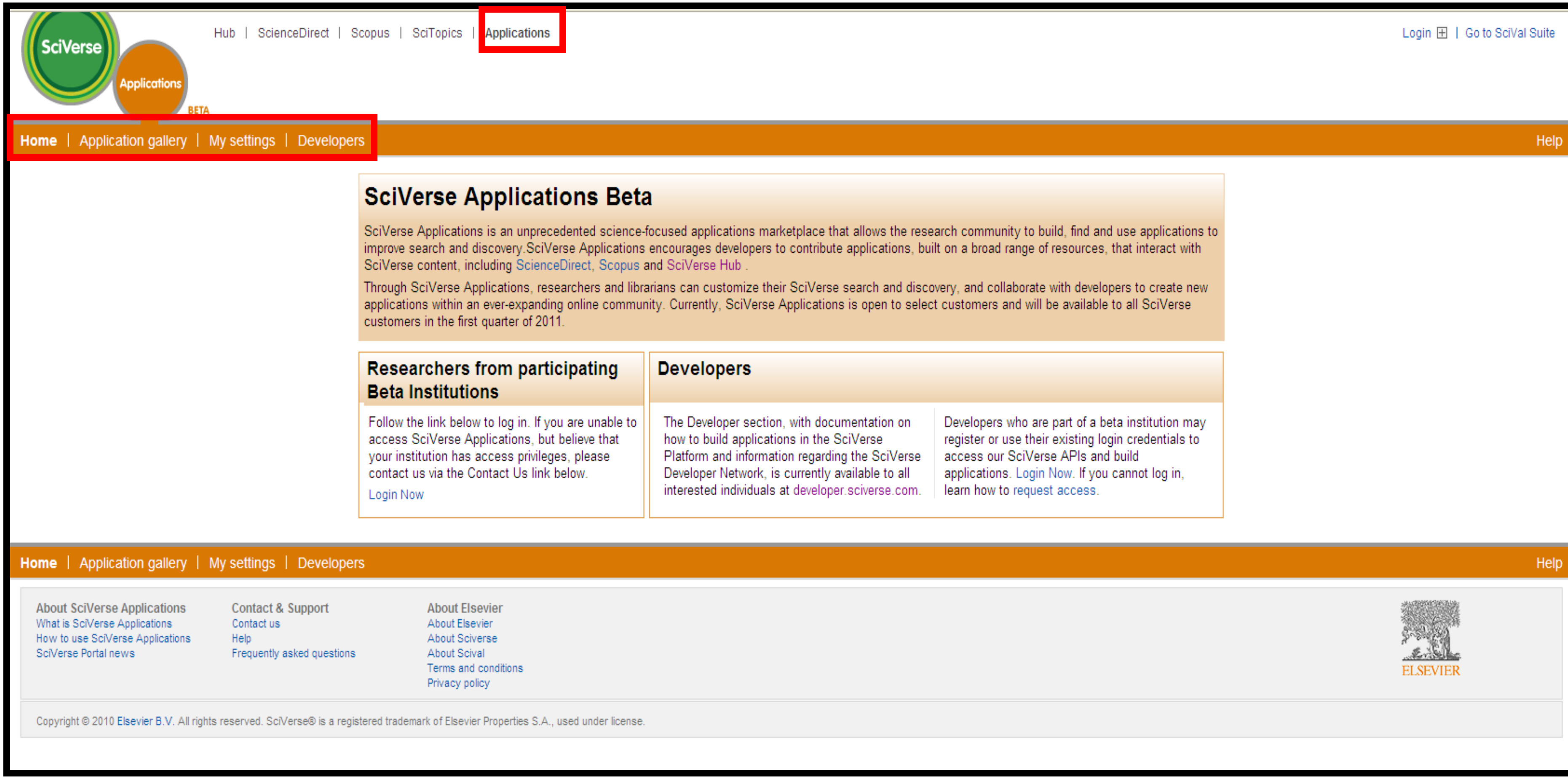

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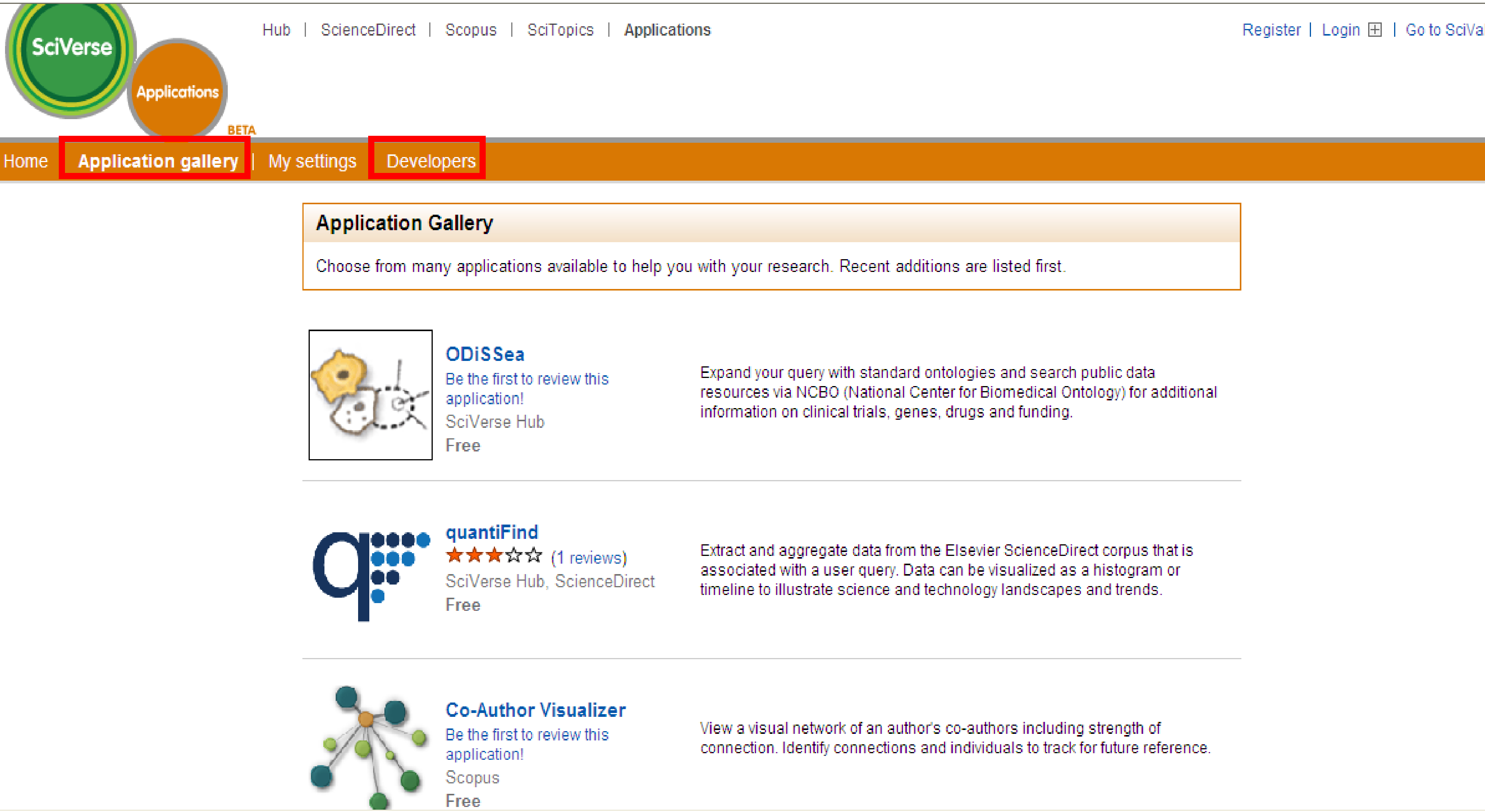

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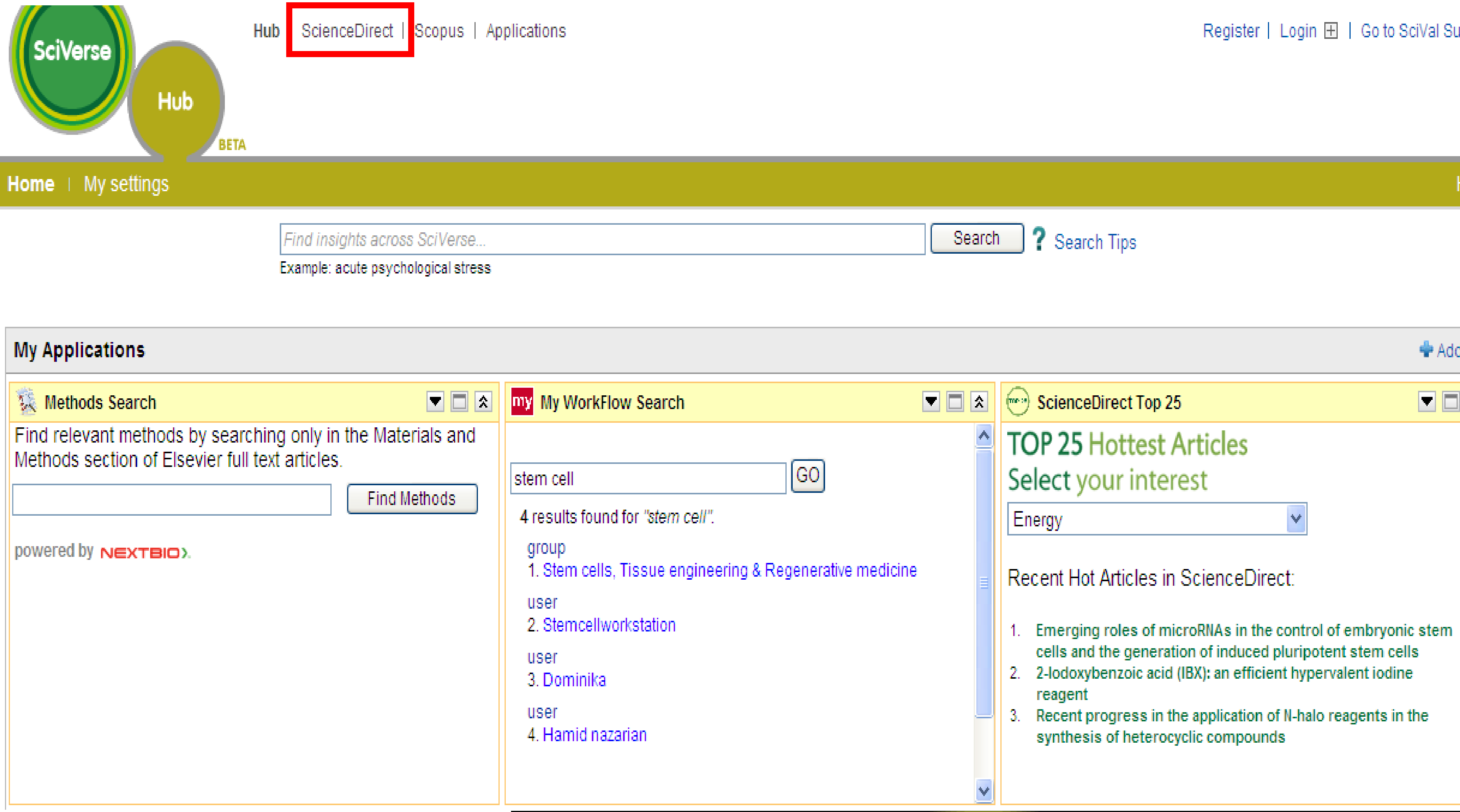

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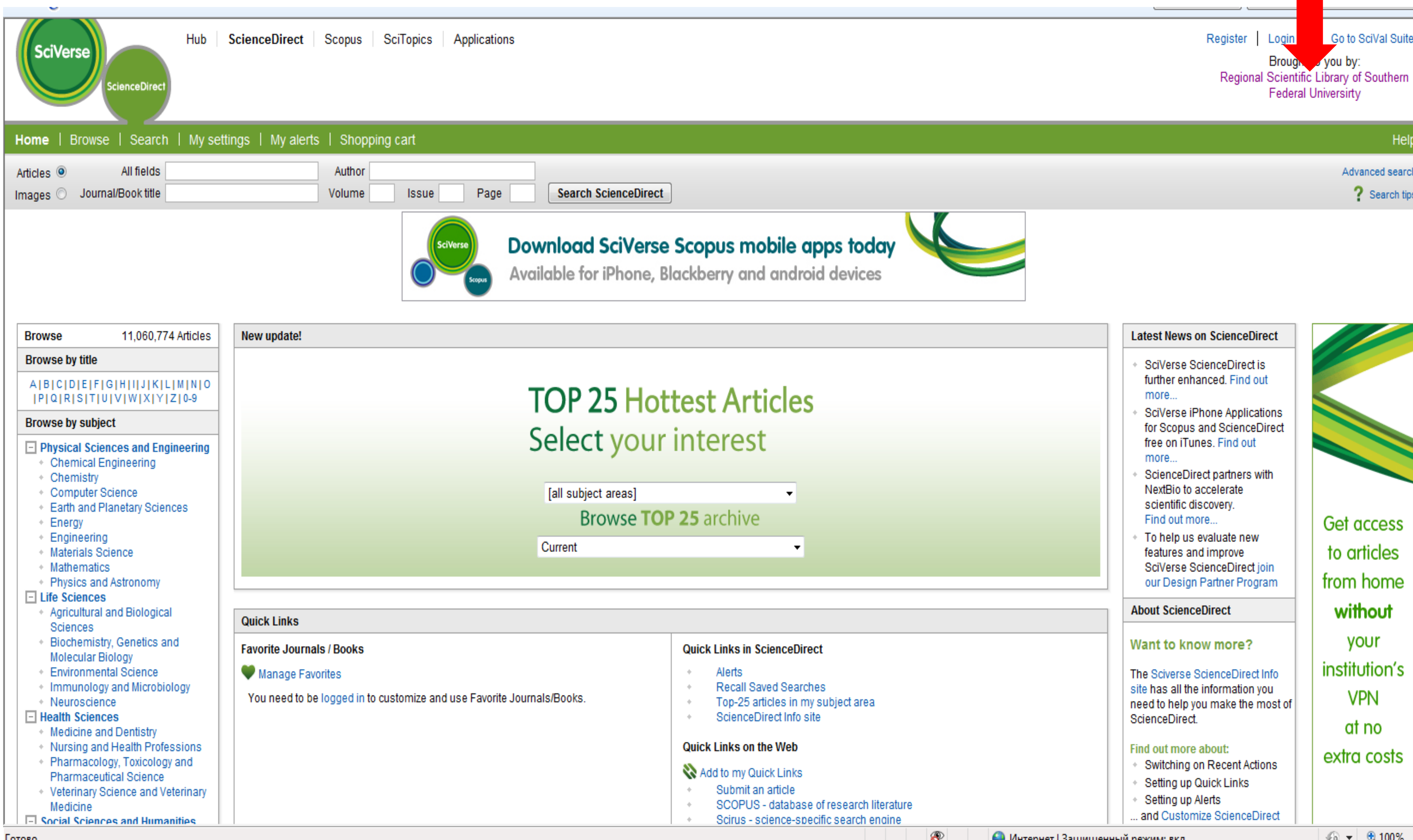

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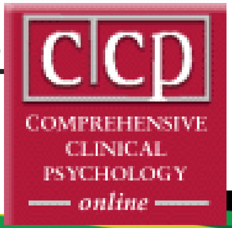

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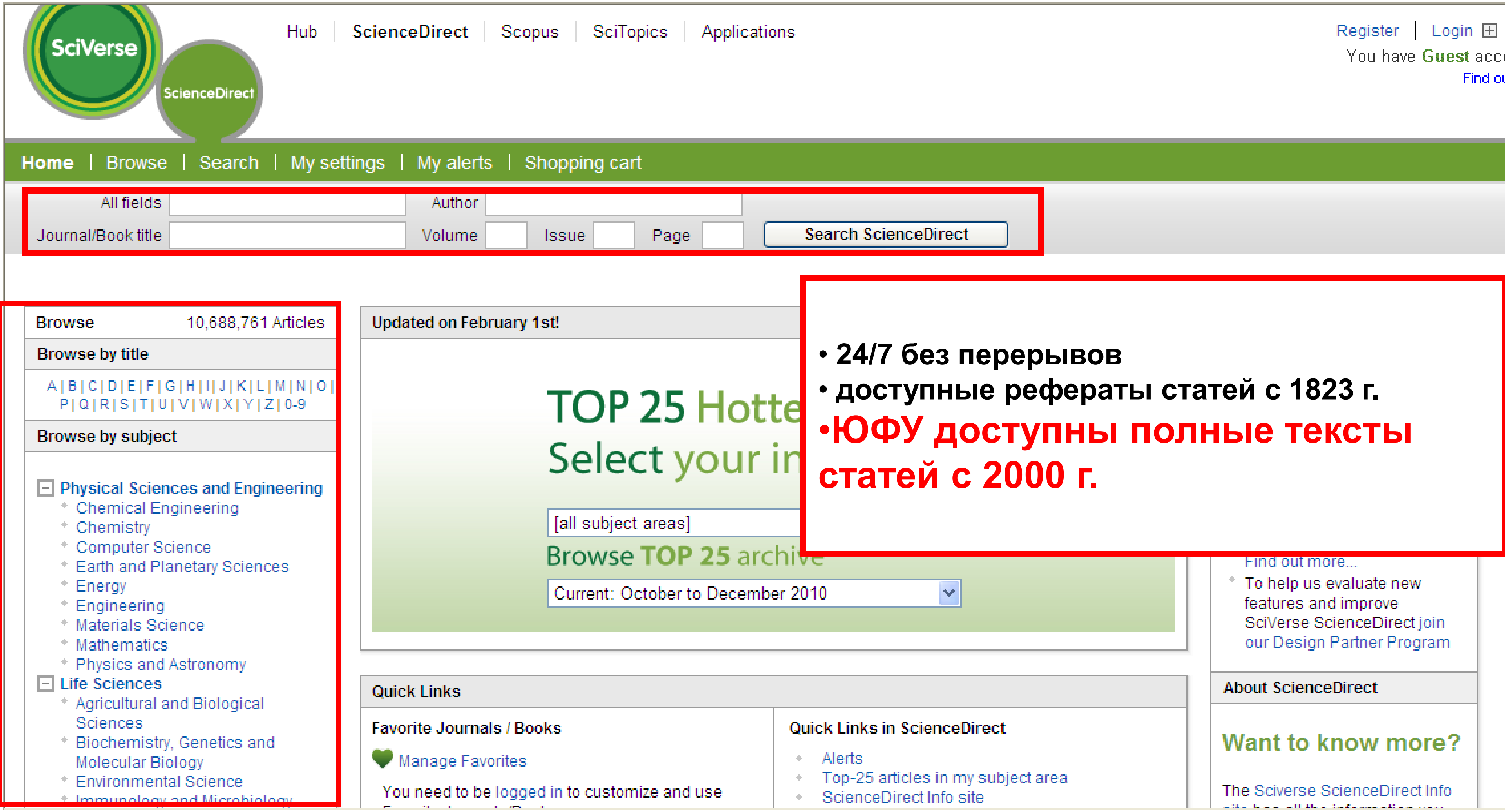

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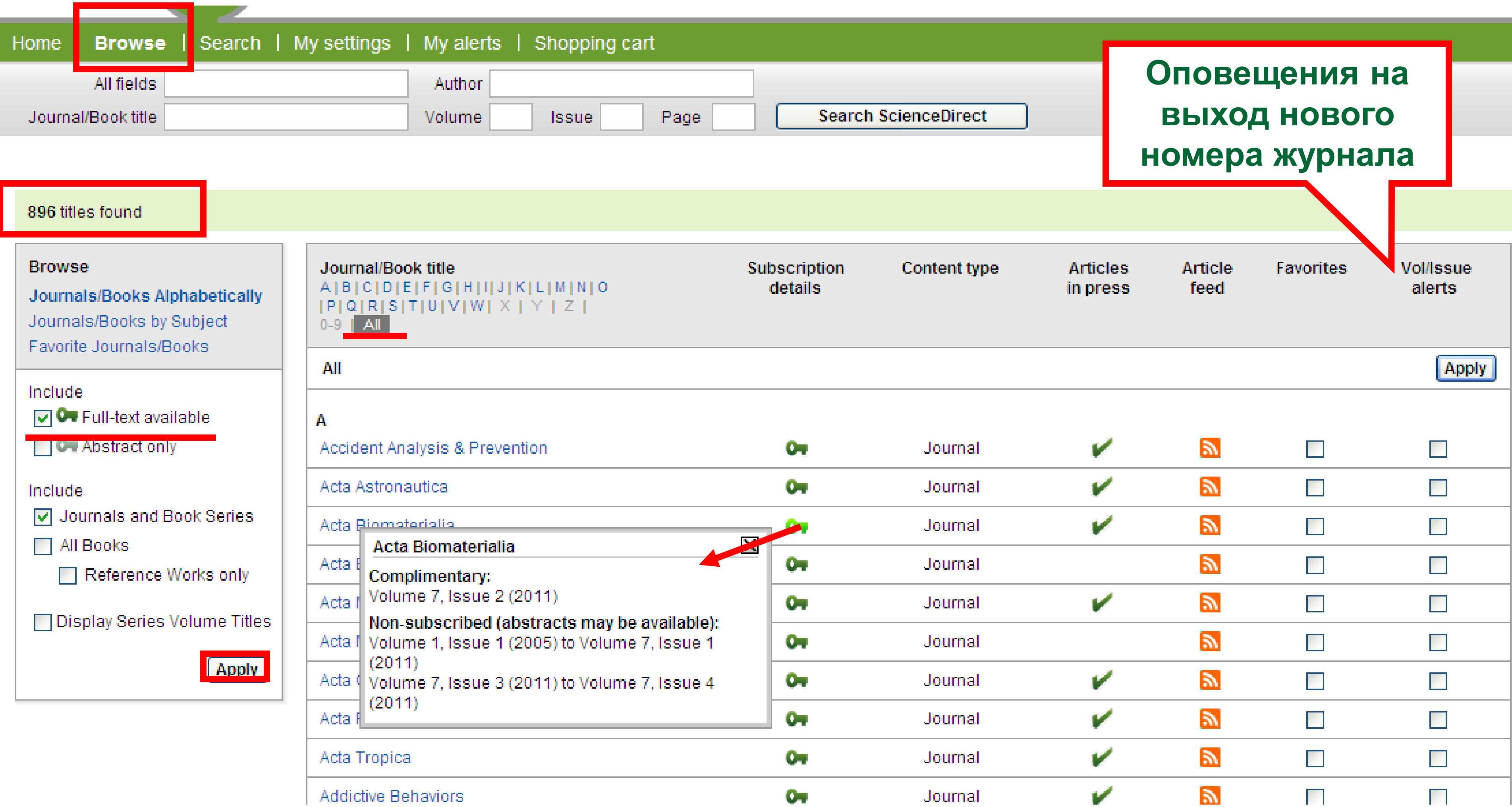

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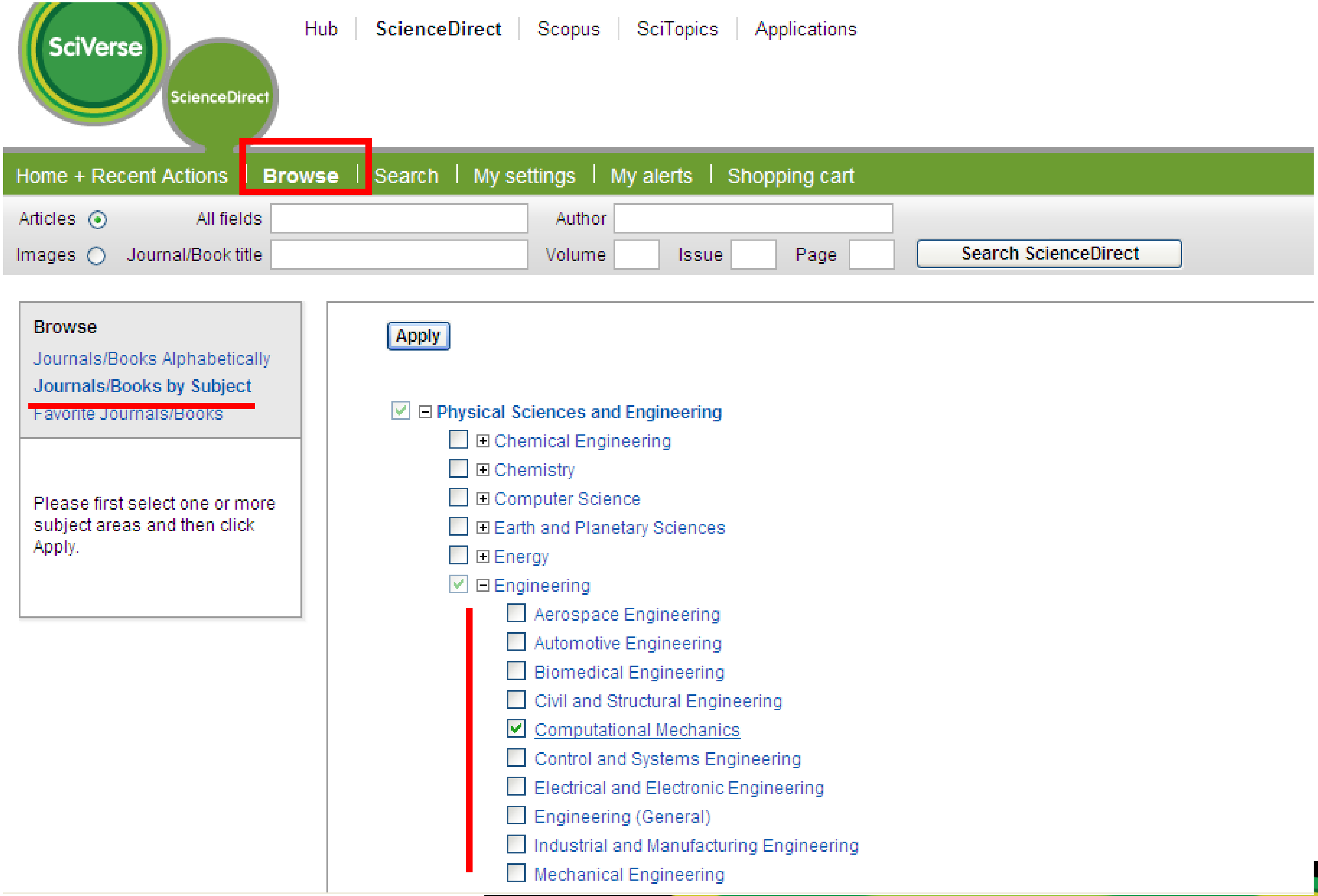

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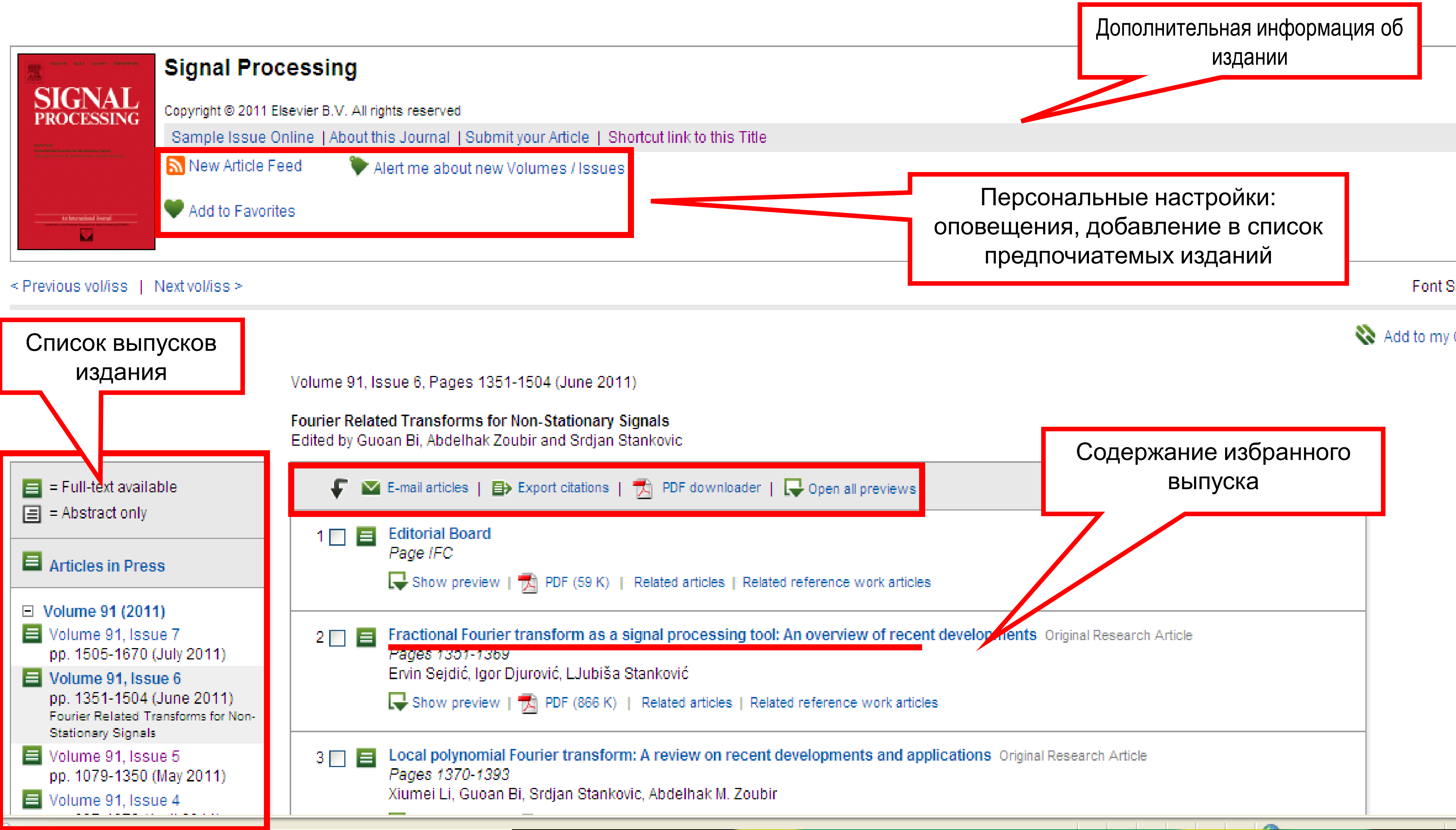

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Thin Solid Films
Volume 487, Issues 1-2, 1 September 2005, Pages 31-34
International Conference on Polycrystalline Semiconductors- Materials, Technologies, Device Applications

doi:10.1016/j.tsf.2005.01.030 | How to Cite or Link Using DOI

Structural and optical properties of $AgIn_5S_8$ films prepared by pulsed laser deposition

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[a]Department of Chemistry, Belarusian State University of Informatics and Radioelectronics, P. Brovka 6, 220027 Minsk, Belarus
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Available online 4 March 2005.

Abstract
$AgIn_5S_8$ thin films were prepared by pulsed laser deposition onto glass substrates. The $AgIn_5S_8$ crystals, as target material, were prepared by Bridgman growth (vertical variant). Crystal structure, chemical composition, surface morphology of laser-deposited films and transmittance and reflectance spectra in the 0.4–2.5-μm wavelength range have been investigated. Band-to-band transition energies were calculated from the optical spectra in the region of the fundamental absorption edge. The obtained results are compared to existing data of single-crystalline $AgIn_5S_8$ bulk material.

Keywords: Depositon process; Composition; Thin Film; X-ray diffraction

Article Outline

1. Introduction

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1. Introduction

$A^{I}B^{III}C^{VI}_2$ (A—Cu,Ag; B—Al,In,Ga; C—S,Se,Te) compounds are promising materials for the fabrication of a variety electro-optical devices and high efficiency thin film solar cells (η≈19%) [1], [2] and [3]. The physical characteristics of these materials can be changed by obtaining solid solutions based on these compounds [1]. However, on cuts of $A^{I}B^{III}C^{VI}$ systems alongside with $A^{I}B^{III}C^{VI}_2$ compounds there are also ordered phases with the general formula $A^{I}B^{III}C^{VI}_8$ [4] and [5] which should also be considered as an effective approach in preparation of materials with required electrical and optical properties. Physical characteristics of such phases are still investigated insufficiently and in this connection their practical potential up to the end is not found yet. The compound $AgIn_5S_8$ belongs to defective semiconductors with a typi[…] around 25%. The electrophysical properties of this compound depend weakly on the introduction[…] under radiating influences that allows to use its as a perspective material for semiconductor and[…] The three elements may form several compounds, depending on the preparation condition. This[…] conditions and an investigation of physical properties of these materials. The excellent performa[…] the preparation of high-temperature superconductors has now become a popular thin film depos[…] materials [6] and [7]. The main advantages of this method are the reproducibility of the target com[…] arrangement.
In this contribution, we have investigated the feasibility of pulsed laser deposition for preparation[…] composition, surface morphologies and optical properties of the films on the glass substrates w[…]

Photoconductivity kinetics in AgIn5S8 thin films

Journal of Alloys and Compounds, Volume 508, Issue 2, 22 October 2010, Pages 380-383

Abstract

The temperature (T) and illumination intensity (F) effects on the photoconductivity of as grown and heat-treated AgIn5S8 thin films has been investigated. At fixed illumination intensity, in the temperature region of 40-300 K, the photocurrent (Iph) of the films was observed to decrease with decreasing temperature.

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2. Experimental details

The $AgIn_5S_8$ crystals, as target material, were prepared by Bridgman growth (vertical variant). The starting components (25 g) from high purity metals (Silver, Indium—99.999% and Sulphur—99.9999%) were taken in proportions corresponding to the stoichiometric composition of the compound and loaded into quartz ampoules with a conical bottom. The growth of crystals was carried out in ampoules about 170 mm in length with diameter between 15–22 mm, that were chemically treated by etching in a mixture of acids HNO_3/HCl=1:3, washing in distilled water and annealing for 2 h at 1270 K. The ampoules were evacuated down to 10^{-3} Pa and sealed. A quartz pivot was welded to the bottom part of the ampoule and then it was

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1270 K. The ampoules were evacuated down to 10^{-3} Pa and sealed. A quartz pivot was welded to the bottom part of the ampoule and then it was placed into a vertical one-zone furnace with a temperature gradient. For the synthesis, the temperature of the furnace was increased from room temperature with the rate of 50 K/h up to 1370 K which is melting point of $AgIn_5S_8$. The melt was homogenized for 3 h by vibration mixing and subsequently directionally crystallized by lowering the furnace temperature with a rate of 2 K/h. An annealing step at 1000 K for 200 h was used for further homogenization of the as-grown ingots. Then the temperature was lowered to 700 K and the furnace was switched off. The crystals obtained by this method had a diameter of 14 mm and a length of around 50 mm.

The fabricated targets were 14 mm in diameter, cut from the as-grown crystals. The $AgIn_5S_8$ thin films were then prepared on Corning glass substrates (2×2 cm^2) by PLD using a Nd:YAG laser operated at a wavelength of 1.06 μm with a pulse duration of 10^{-3} s. Substrate temperatures (T_S) in the range 450–480 °C were used with a target-to-substrate distance of 7–10 cm. Laser pulses (20–40) with a repetition rate of 3×10^{-2} Hz were incident on the target at an angle of 45°. The partly focused laser (focal length=500 mm) beam was scanned over the target surface and the films were deposited at a residual pressure of about 10^{-5} Torr. By changing the high voltage between 2.3 and 2.5 kV of the laser, the laser energy was varied from 150 to 180 J per pulse. The laser spot size on the target was measured to be 3×5 mm^2. Typical deposition rates were $(2–4)\times10^5$ Å/s and the thickness of the films was 700–1500 nm, measured with a surface profile measuring apparatus.

The phases and crystallographic structure of the $AgIn_5S_8$ crystals and films were investigated by X-ray diffraction (XRD) using Cu K_α (λ=1.5405 Å) radiation in the range 2Θ=15–100°. The observed phases were determined by comparing the *d*-spacing with Joint Committee on Powder Diffraction Standard (JCPDS) data files. Bulk compositions and surface morphologies of the targets and films were investigated by electron probe microanalysis (EPMA) using a CAMECA SX-100 and by energy-dispersive analysis of X-ray measurements (EDX) carried out by the use of a JEOL 6400 SEM apparatus. The depth profiling was done by Auger electron spectroscopy (AES) using a Perkin Elmer Physical Electronics model 590. An Ar ion beam was used for sputter etching. The transmittance as well as reflectivity measurements were carried out at room temperature using a "Becman-520" and "Specord-61NIR" spectrometers in the spectral range from 500 to 2500 nm. This allowed to determine the absorption coefficients and energy gaps of the films.

Results and discussion

X-ray diffraction measurements were performed to identify the structure and phases in the as-grown films. The diffraction spectra of the powder target materials were also recorded for comparison. Before the X-ray analysis, the crystals were powdered, pressed into a special holder and annealed in vacuum at 650 K for 2 h in order to remove the mechanical stresses. Fig. 1 shows diffractograms of $AgIn_5S_8$ crystals and thin films. The resolution of high-angle reflections shows that the crystals and thin films are single phase. No extra phases appeared, confirming the adequacy of our methods as regards the good phase quality of the synthesized materials. The X-ray powder diffraction patterns of crystals and film samples showed a set of lines due to the spinel structure (O^7_h-*Fd*3*m* space group). The spectra demonstrate that the films always grew with a preferred $\langle 111 \rangle$ orientation.

Based on the measured values of diffraction angles, the *d*-spacings for various planes of the reflection have been calculated. The lattice parameters

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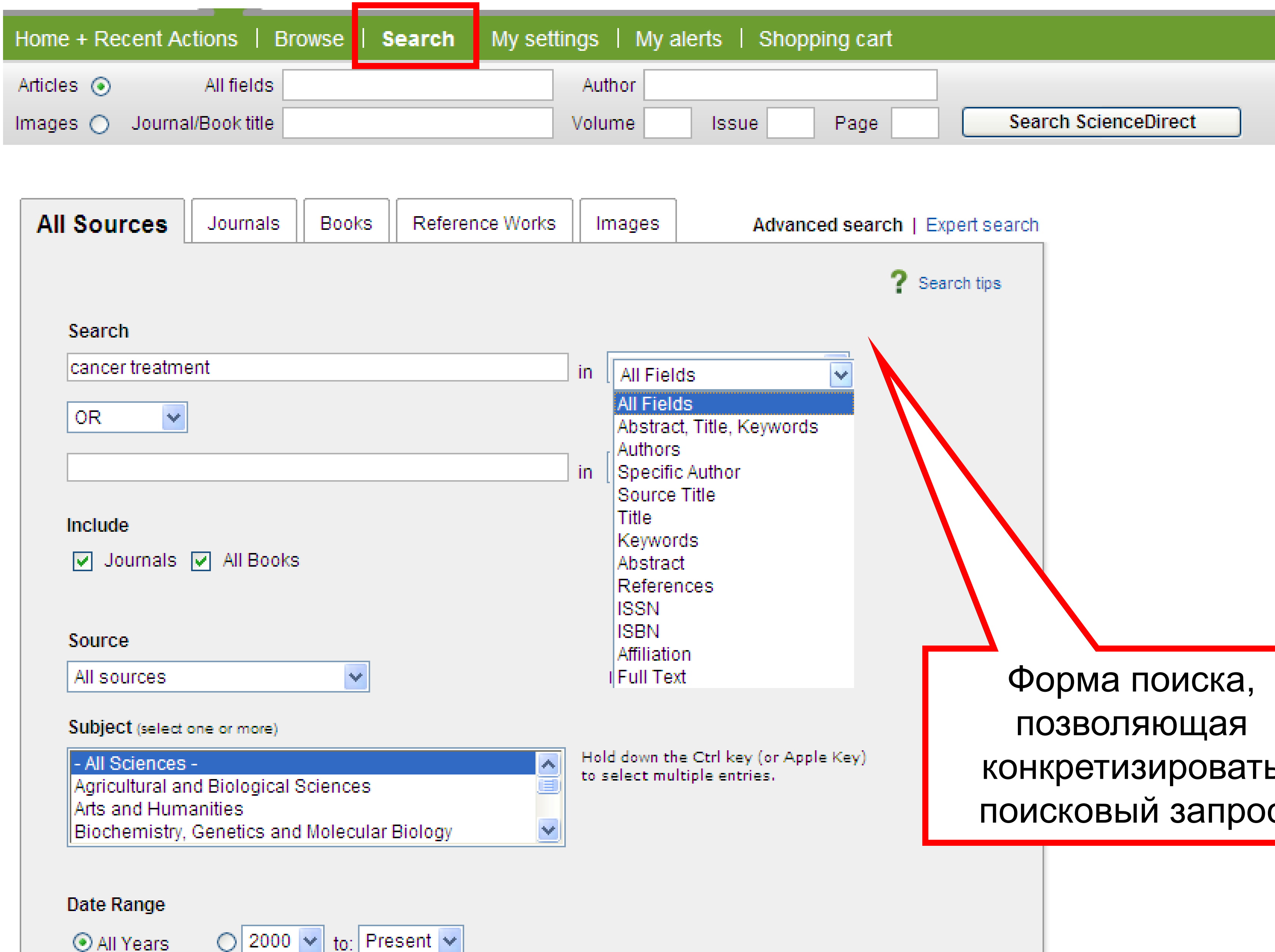

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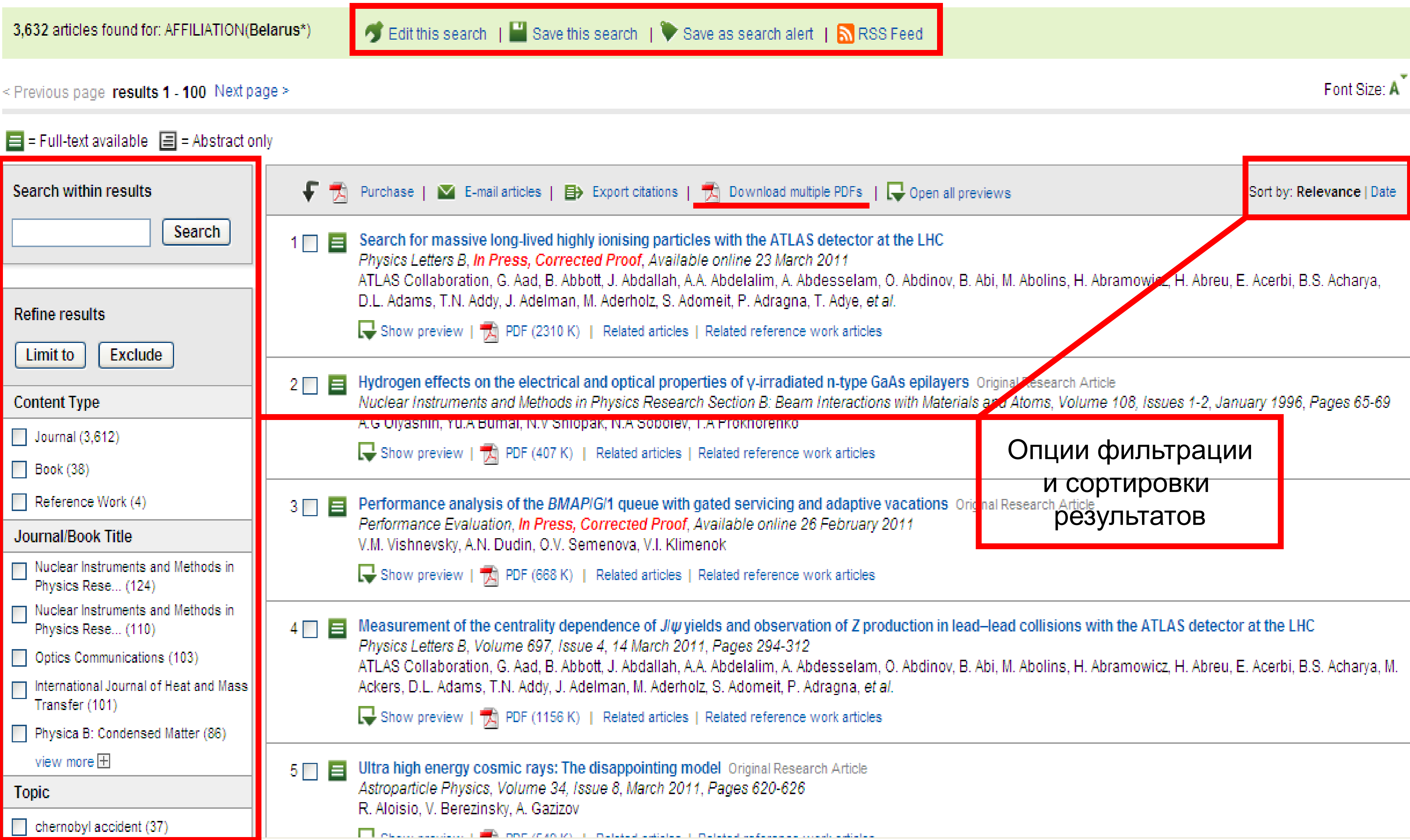
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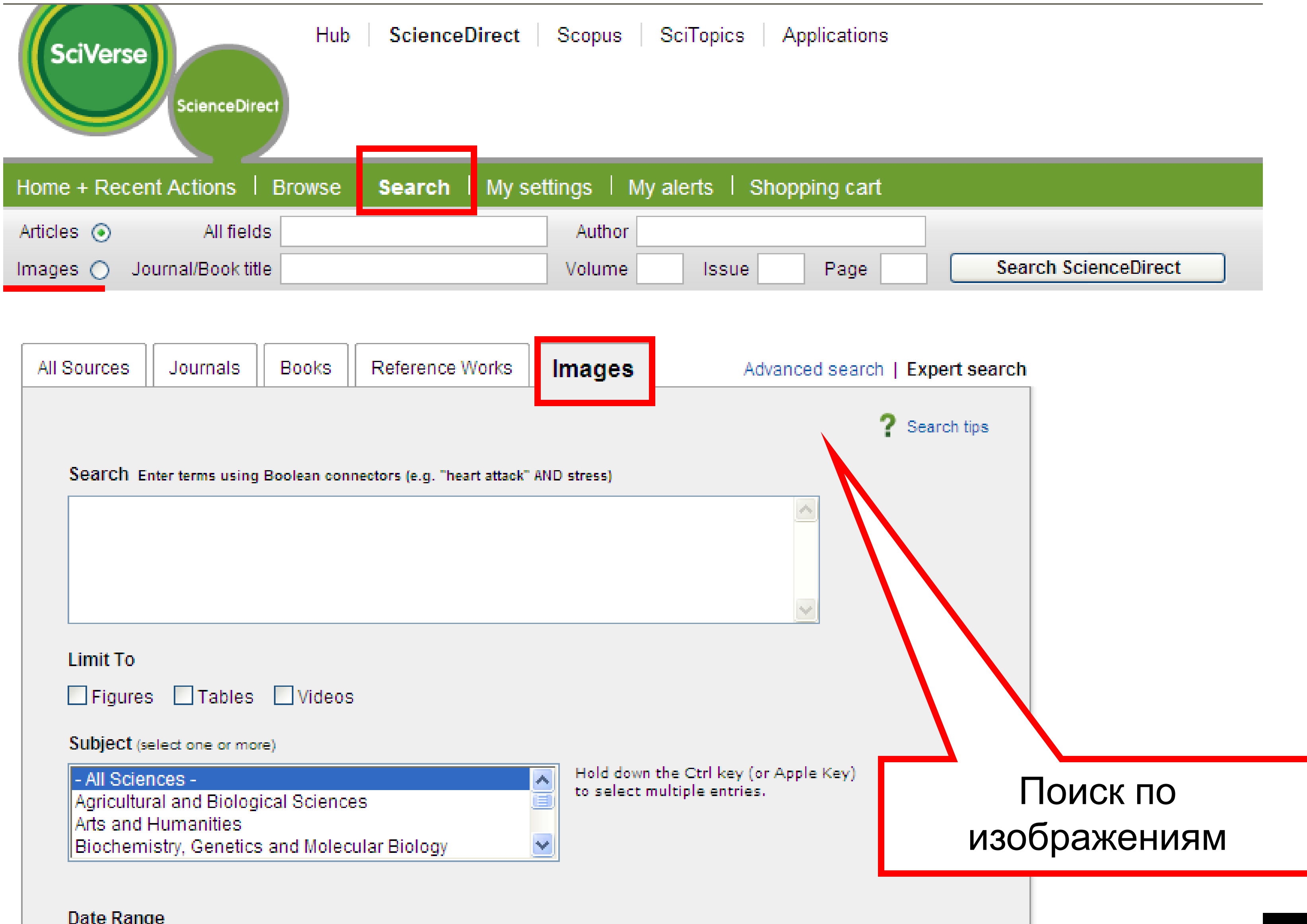

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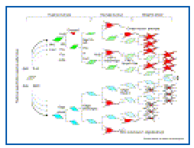

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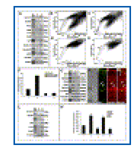

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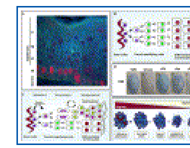

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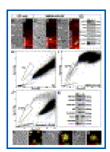

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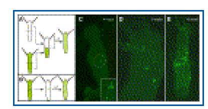

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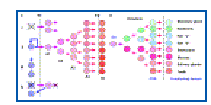

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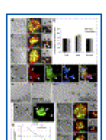

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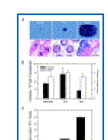

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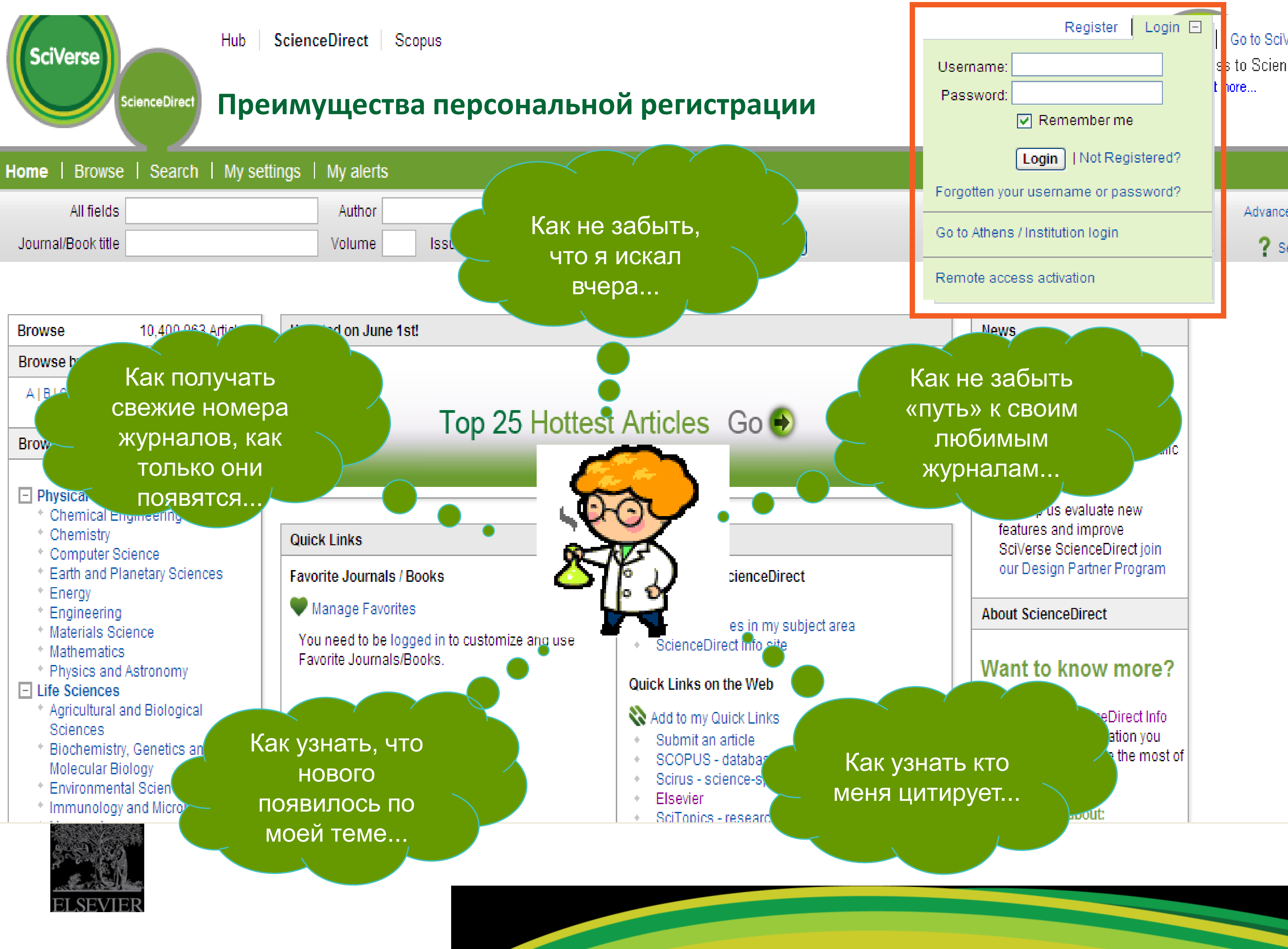

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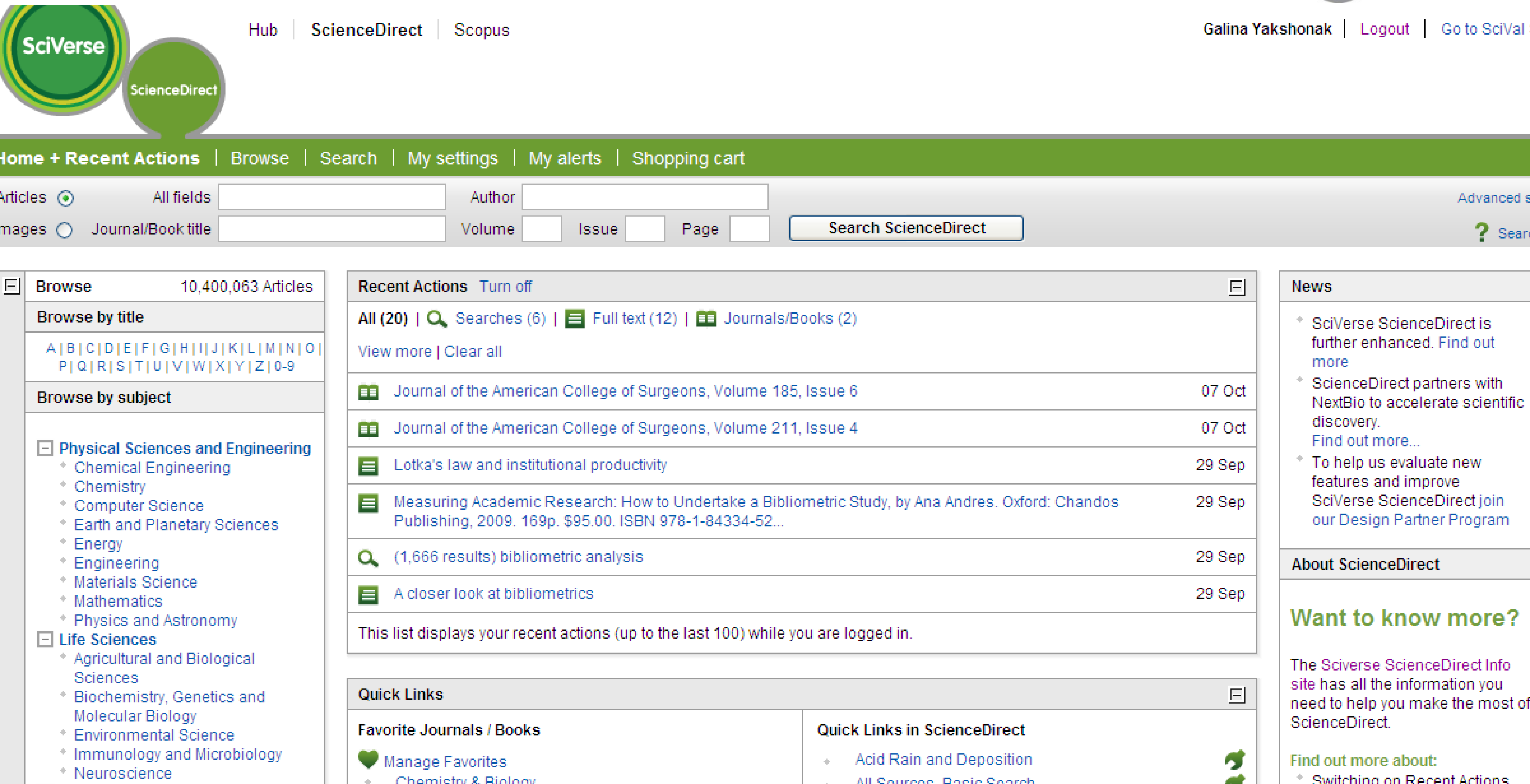

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